# AI-Co-Pd (Aluminum-Cobalt-Palladium)

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Recent investigations of this system [2000Yur, 2002Yur] have revealed the existence of a number of ternary compounds in the Al-rich region. A comparative study of the Al-Co-Pd and Al-Co-Ni systems was presented by [2004Yur].

## **Binary Systems**

The Al-Co phase diagram [1996God] shows the following intermediate phases: CoAl (48-78.5 at.% Co; B2, CsCltype cubic), Co<sub>2</sub>Al<sub>5</sub> (D8<sub>11</sub>-type hexagonal), CoAl<sub>3</sub> (D0<sub>11</sub>, Fe<sub>3</sub>C-type orthorhombic), Co<sub>4</sub>Al<sub>13</sub> (three modifications with one orthorhombic and two monoclinic forms, denoted  $Co_4Al_{13}(o)$ ,  $Co_4Al_{13}(m)$ , and Z, respectively by [2002Yur]), and  $Co_2Al_9$  (D8<sub>d</sub>-type monoclinic). The Al-Pd phase diagram [2001Yur] depicts the following intermediate phases: PdAl<sub>4</sub> (hexagonal, space group P6<sub>3</sub>22), PdAl<sub>3</sub> (denoted  $\varepsilon_6$ , orthorhombic),  $\varepsilon_{28}$  (~PdAl<sub>3</sub>, orthorhombic),  $Pd_8Al_{21}$  ( $Pt_8Al_{21}$ -type tetragonal),  $Pd_2Al_3$  (denoted  $\delta$ ,  $D5_{13}$ , Ni<sub>2</sub>Al<sub>3</sub>-type hexagonal), PdAl (B2-type cubic and two lowtemperature forms: rhombohedral and B20-type cubic), Pd<sub>5</sub>Al<sub>3</sub> (Rh<sub>5</sub>Ge<sub>3</sub>-type orthorhombic), Pd<sub>2</sub>Al(C23, Co<sub>2</sub>Sitype orthorhombic), and Pd<sub>5</sub>Al<sub>2</sub> (Pd<sub>5</sub>Ga<sub>2</sub>-type orthorhombic). In the Co-Pd system, Co and Pd form a continuous face-centered cubic (fcc) solid solution.

## The Ternary Phase Equilibria

The structural characteristics of six stable ternary phases of this system were summarized by [2002Yur], see Table 1. Following [2002Yur], these are denoted using the symbols W, V, U, F, C<sub>2</sub>, and Y<sub>2</sub>. In addition, the  $\varepsilon$ -family of

orthorhombic phases  $\varepsilon_{22}$  and  $\varepsilon_{34}$  were identified by [2002Yur] in the ternary region, Table 1. All  $\varepsilon$ -type phases are clubbed together and labeled  $\varepsilon$  in the isothermal sections drawn by [2002Yur].

With starting metals of 99.9999% Al, 99.95% Co, and 99.95% Pd, [2002Yur] levitation-melted 90 Al-rich ternary alloys. The alloys were annealed at 1050-790 °C for 100-2050 h. The phase equilibria were studied by x-ray powder diffraction and scanning electron microscopy. Local phase compositions were measured with energy dispersive x-ray analysis and inductively-coupled plasma optical emission



Fig. 1 Al-Co-Pd isothermal section at 1050 °C [2002Yur]

Table 1	Al-Co-Pd	Crystal	Structure an	d Lattice	Parame-ter	Data	[2002Yur]	
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			Lattice parameter, nm				
Phase designation	Composition, at.%	Space group	а	b	с	β(°)	
W	72.0 Al 23.2 Co 4.8 Pd	$Pmn2_1$	2.36	0.82	2.07		
V	70.0 Al 20.0 Co 10.0 Pd	P121, P1m1 or P12/m1	1.0068	0.3755	0.6512	102.38	
U	69.1 Al 14.4 Co 16.5 Pd	C121, C1m1 or C12/m1	1.9024	2.9000	1.3140	117.26	
F	72.8 Al 18.2 Co 9.0 Pd	$P2_1/a\overline{3}$	2.4397				
C <sub>2</sub>	63.0 Al 10.5 Co 26.5 Pd	$Fm\overline{3}$	1.5507				
Y <sub>2</sub>	75.5 Al 20.7 Co 3.8 Pd	Immm	1.5451	1.2105	0.7590		
ε <sub>22</sub>	72.0 Al 8.0 Co 20.0 Pd	(a)	2.35	1.68	5.70		
ε <sub>34</sub>	70.0 Al 15.0 Co 15.0 Pd	(a)	2.35	1.68	7.01		
(a) Orthorhombic							



Fig. 2 Al-Co-Pd isothermal section at 1000 °C [2002Yur]



Fig. 3 Al-Co-Pd isothermal section at 940 °C [2002Yur]



Fig. 4 Al-Co-Pd isothermal section at 790 °C [2002Yur]

spectroscopy. Four partial isothermal sections were constructed in the Al-rich region at 1050, 1000, 940, and 790 °C, as shown in Fig. 1-4.

At 1050 °C (Fig. 1), CoAl and PdAl form a continuous *B*2 solid solution. The maximum solubility of Al in *B*2 is 58.5 at.% and occurs at 18 at.% Pd. The solubility of Pd in Co<sub>2</sub>Al<sub>5</sub>, Co<sub>4</sub>Al<sub>13</sub>(m) and Z phases is 3.0, 2.7, and 3.4 at.%, respectively. The ternary phases F, V, and W are present and form tie-lines with the liquid. The W phase has a small homogeneity range around the composition Al<sub>72.5</sub>Pd<sub>4</sub>Co<sub>23.5</sub>. The Pd content of the F phase is 9.4-8.2 at.% at approximately constant Al content. The V phase has a composition range of Al<sub>69.7</sub>Pd<sub>9.5</sub>Co<sub>20.8</sub>-Al<sub>71</sub>Pd<sub>8.6</sub>Co<sub>20.4</sub>. The binary phases Co<sub>2</sub>Al<sub>9</sub> and Pd<sub>2</sub>Al<sub>3</sub> are not stable at this temperature.

At 1000 °C (Fig. 2), the Al solubility in *B*2 remains the same as at 1050 °C. The solubility of Pd in Co<sub>2</sub>Al<sub>5</sub>, Co<sub>4</sub>Al<sub>13</sub>(m) and Z phases is 3.1, 3.8, and 2.5 at.%, respectively. The binary phase Pd<sub>2</sub>Al<sub>3</sub> (denoted  $\delta$ ) nucleates in the ternary region and is stable between 7.1 and 3.8 at.% Co. The  $\varepsilon$  phase has a range of 73.2-74.6 at.% Al and 9.6-14 at.% Pd. Among the ternary phases, W is not present, having decomposed between 1050 and 1000 ° C. The F phase has a range of 70.1-71 at.% Al and 8.5-9.8 at.% Pd. Both F and V are no longer in equilibrium with the liquid. The U phase has formed, with a range of 69.4-70.4 at.% Al and 11.3-16.9 at.% Co.

At 940 °C (Fig. 3), the binary phase  $Co_2Al_9$  is additionally present. The  $Pd_2Al_3$  ( $\delta$ ) has spread to the Al-Pd side.

The solubility of Pd in Co<sub>2</sub>Al<sub>9</sub>, Co<sub>4</sub>Al<sub>13</sub>(o), Co<sub>4</sub>Al<sub>13</sub>(m), Z, and Co<sub>2</sub>Al<sub>5</sub> is 0.9, 1.2, 2.7, 1.6, and 3.0 at.%, respectively. The ternary phase V is unstable. The Y<sub>2</sub> phase has become stable around the composition Al<sub>74.8</sub>Pd<sub>3.8</sub>Co<sub>21.4</sub>. The F phase has a range of 8.7-12.4 at.% Pd and 72-72.5 at.% Al. At 790 °C (Fig. 4), the C<sub>2</sub> phase has appeared as an additional ternary phase. The measured compositions of the phases in the three-phase equilibria at each of the above temperatures were listed by [2002Yur].

#### References

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